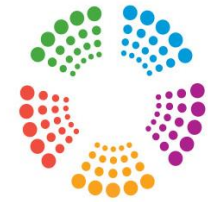
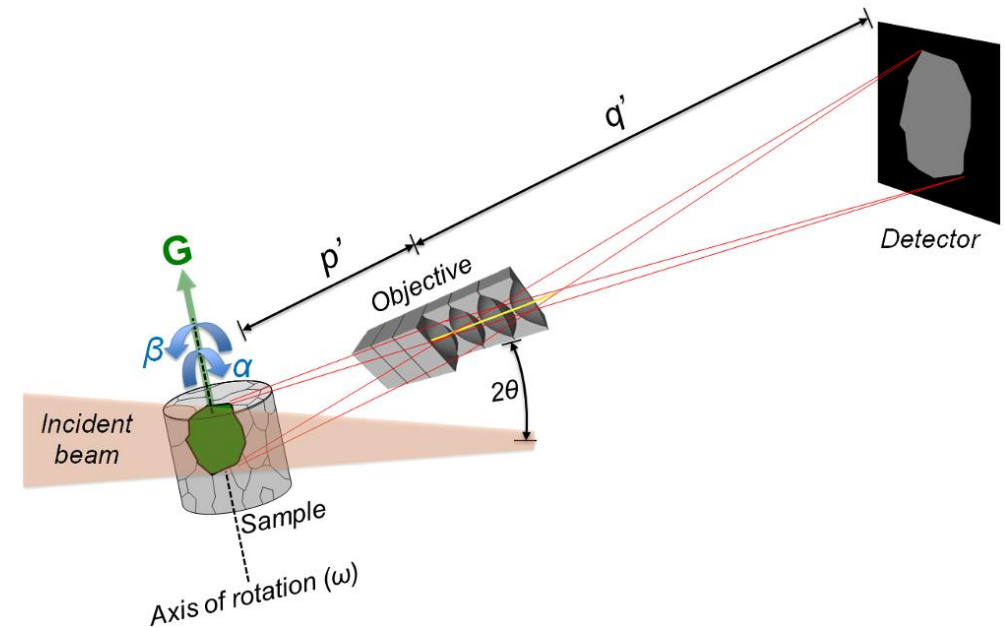


ID03 is the new ID06-HXM, the ESRF Dark-Field X-ray microscopy beamline



STREAMLINE

C. Detlefs (ESRF), on behalf of the ID03 team:
Helena Isern
Raquel Rodriguez Lamas
Thierry Brochard
Thomas Dufrane
Can Yildirim
and many collaborators



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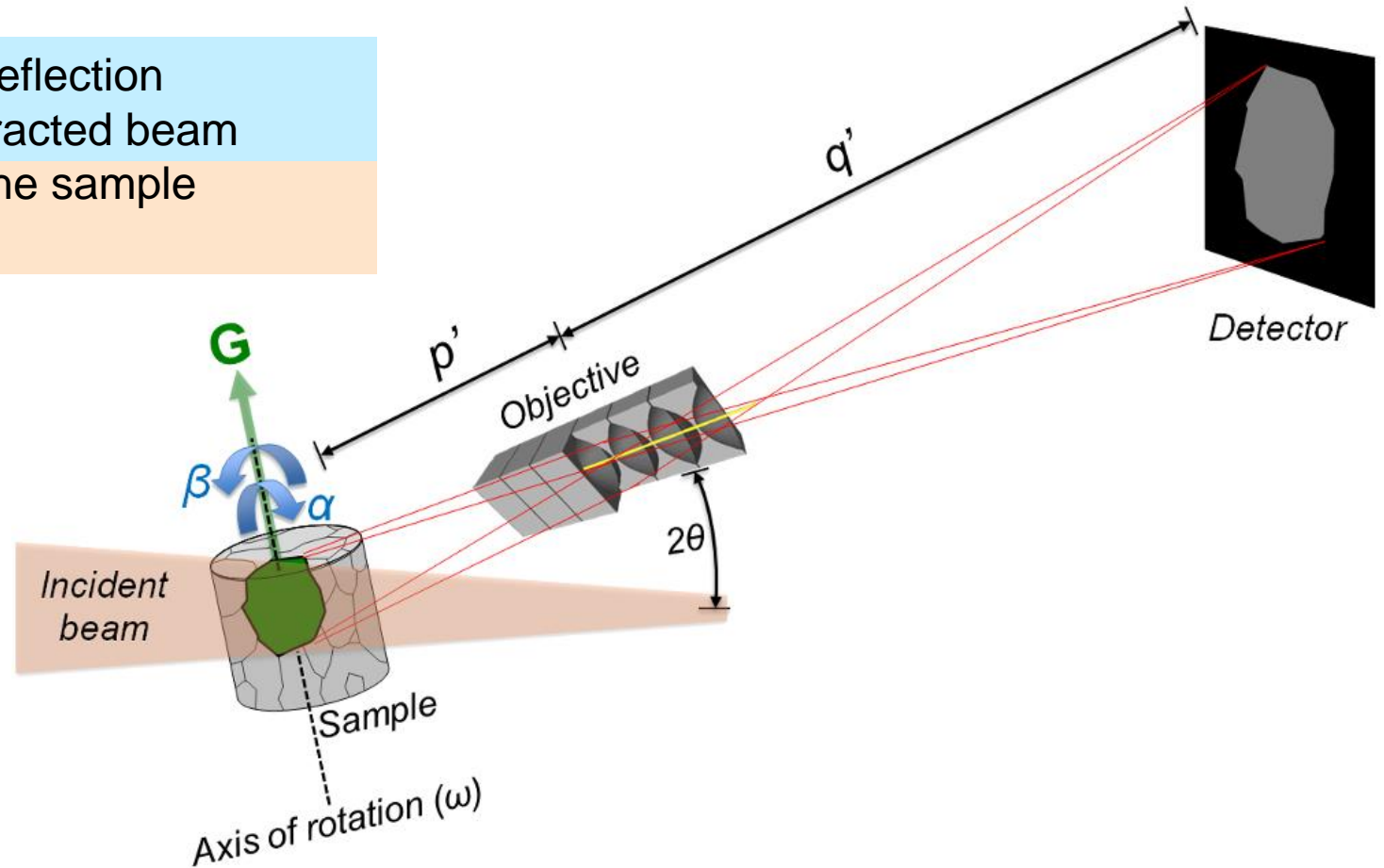
DOI:10.1557/mrs.2020.89

Magnified diffraction topography using a lens between sample and detector

- You take a sample and align a Bragg reflection
- You put an imaging detector in the diffracted beam
- You insert an objective lens between the sample and the detector

- Diffraction topography
- Rocking curve imaging

- X-ray microscopy

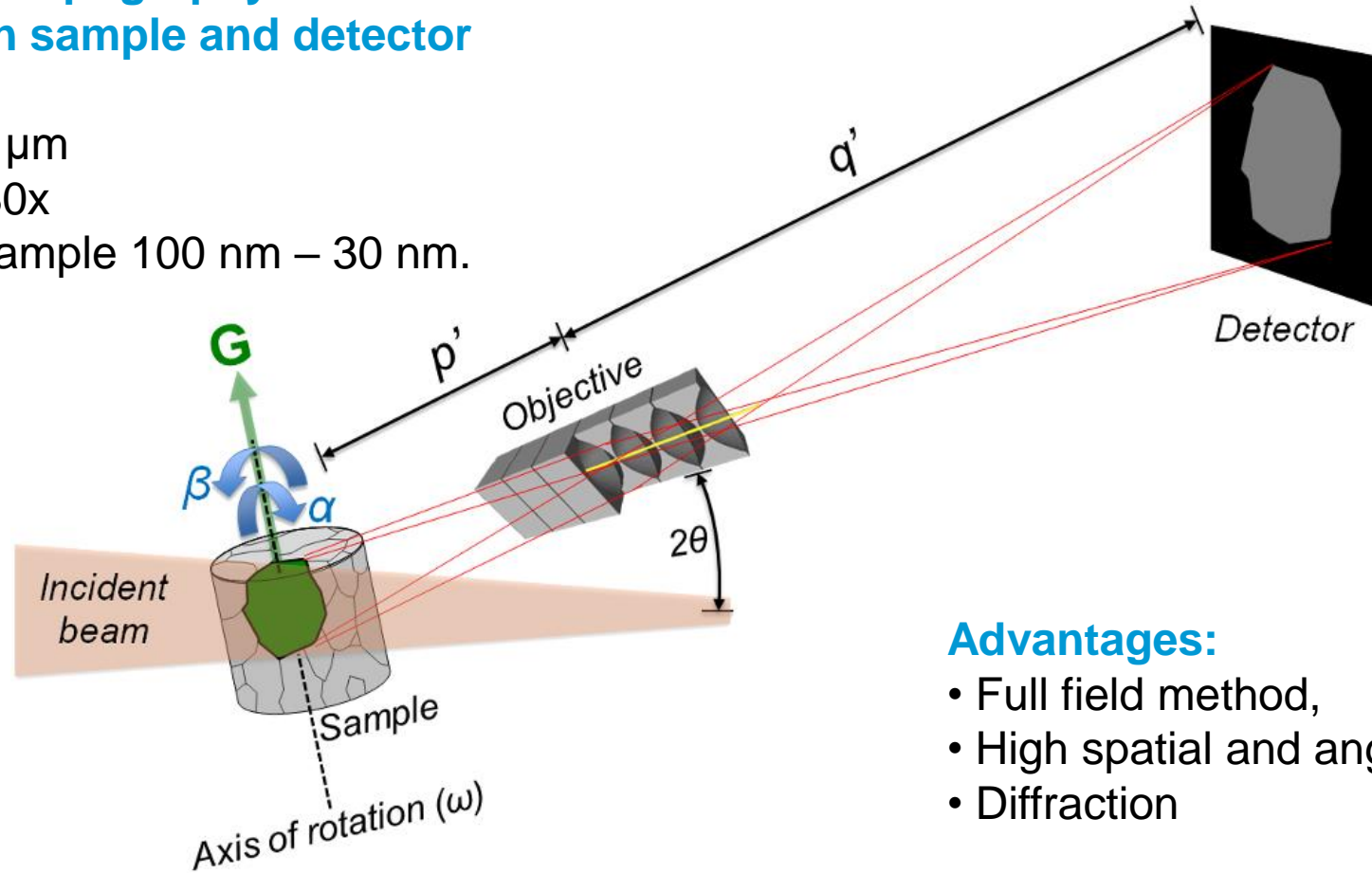


Magnified diffraction topography using a lens between sample and detector

Detector resolution $1\ \mu\text{m}$

Magnification $10\times - 30\times$

→ Resolution at the sample $100\ \text{nm} - 30\ \text{nm}$.



Advantages:

- Full field method,
- High spatial and angular resolution,
- Diffraction

Complementary to electron microscopy (TEM, STEM)

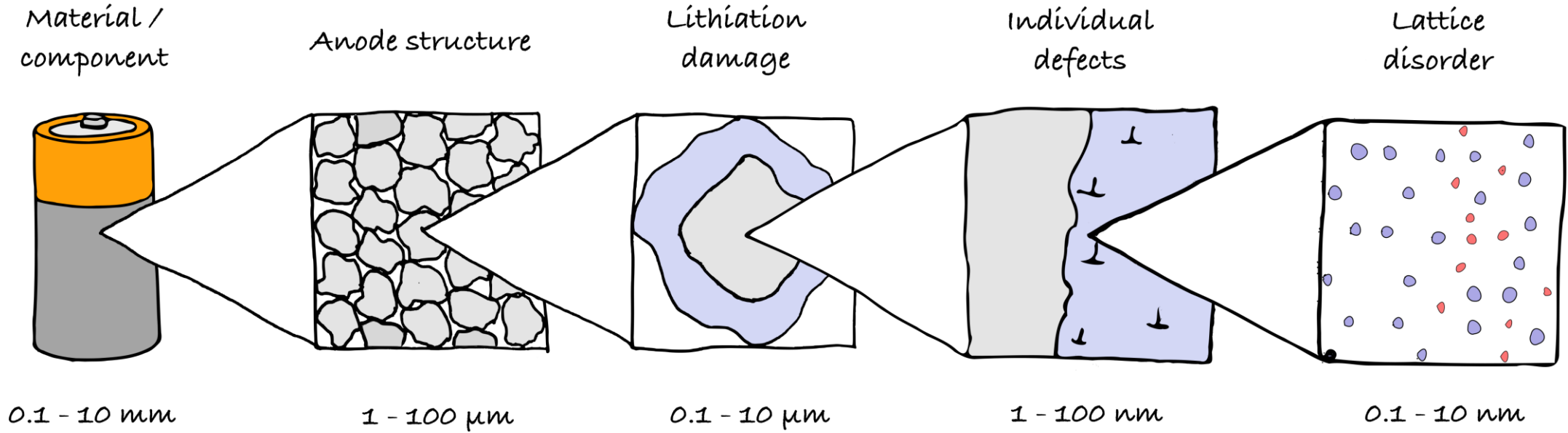
→ both are diffraction based full-field imaging, offering crystallographic sensitivity: phase, orientation, strain, ...

	DXFM	TEM/STEM
Spatial resolution	~150 nm 😞	~1 pm 😊
Sample preparation	<0.5 mm thick, no polishing needed 😊	~100 nm thick, polished 😞
Strain resolution	~0.001% 😊	~0.1% 😞
Sample environment	In-air, possibility of <i>in-situ</i> setups 😊	Vacuum, limited sample environments 😞

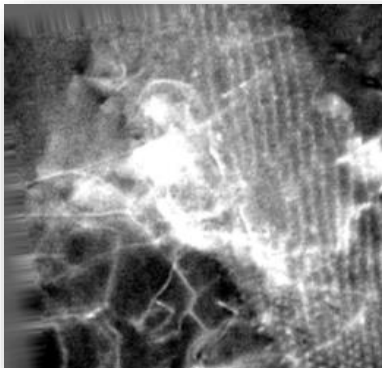
- **Main advantage, due to bulk samples sizes, is the capability to follow sample evolution *in-situ* during processing (temperature, electric field, mechanical strain...)**

Need for Multiscale Imaging

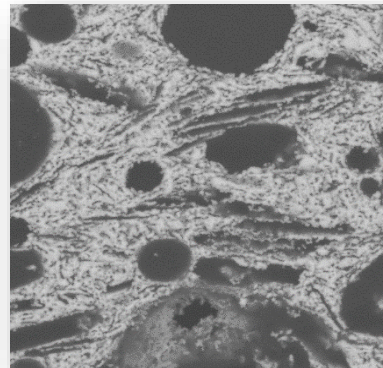
Materials properties are determined by hierarchically organized structures



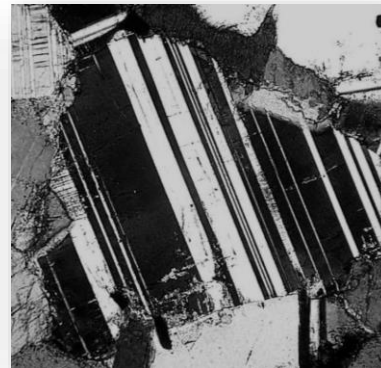
Metals



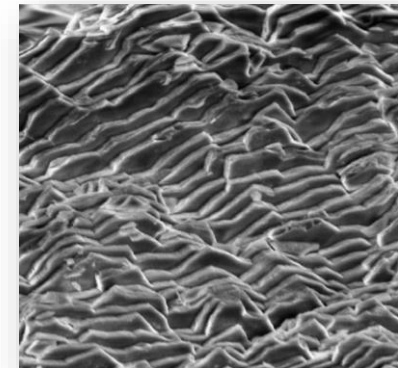
Ceramics



Minerals



Biomaterials



Diffraction technique:

Bragg's law

$$\lambda = 2d \sin \theta$$

Sensitive to variations in the crystal lattice

- Phase
- Orientation of lattice planes
- d -spacing

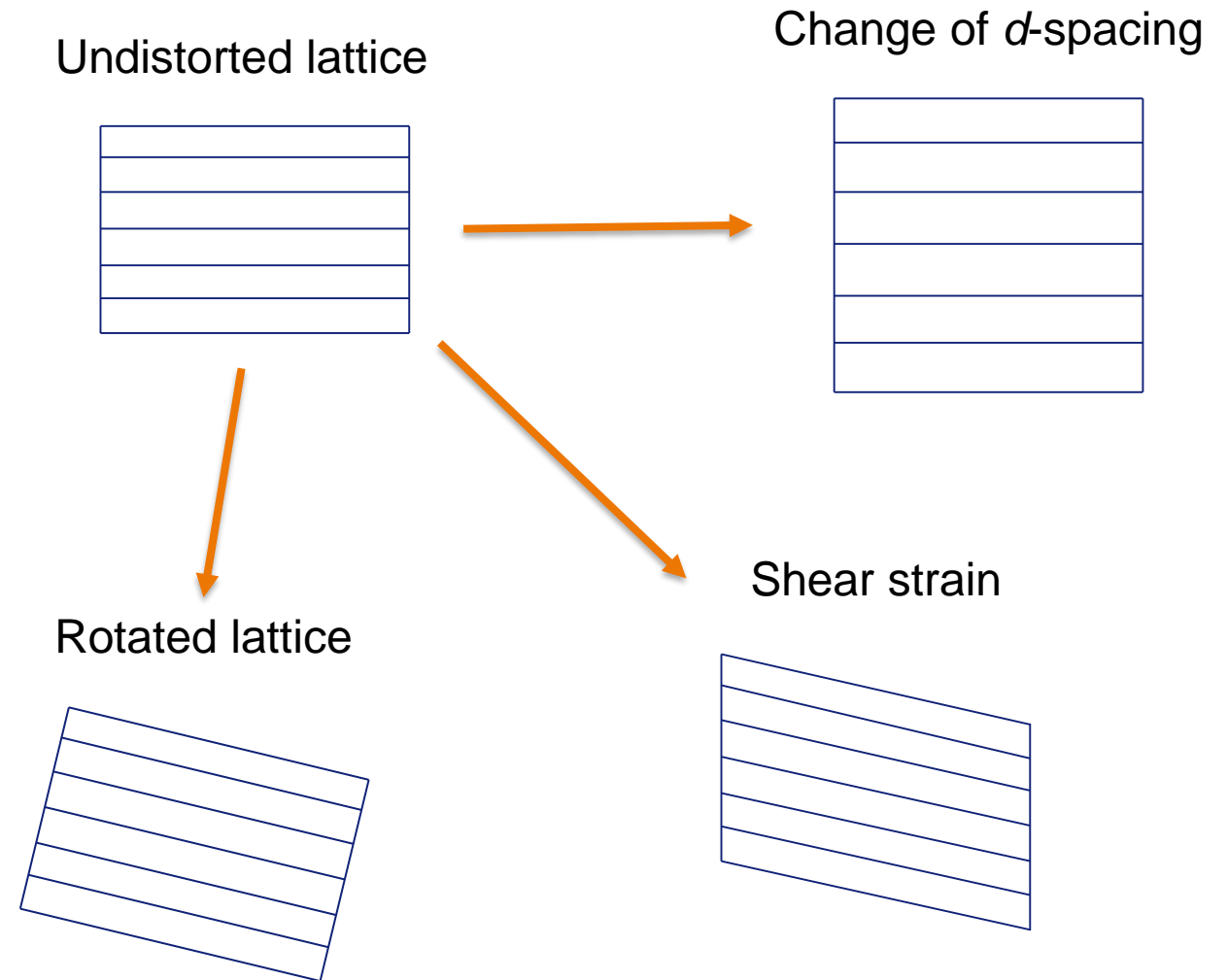
→ Strain

→ Defects such as dislocations

→ Grain/domain boundaries

Measure diffracted intensity as function of crystal orientation and scattering angle

Record a microscopy image at each angle



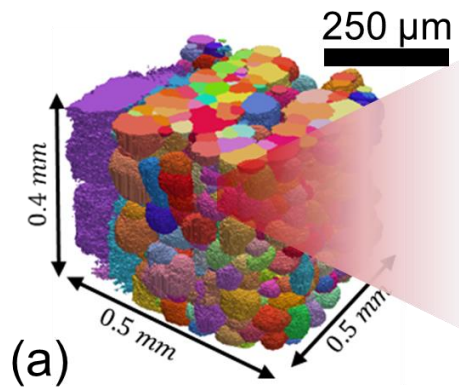
Dark Field X-ray Microscopy

A unique microscope to image embedded structures

- Mapping grain level 3D Lattice strain and orientation
- Can be coupled with 3DXRD/DCT
- Analysis package : *darfix*

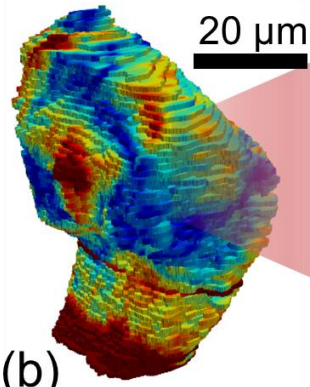


Grains in Polycrystals



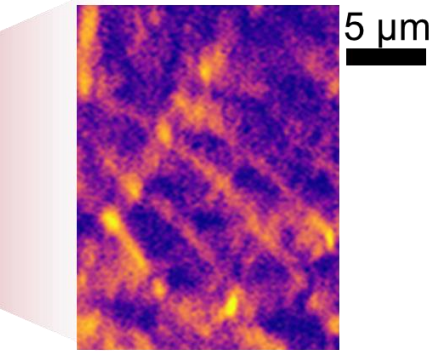
(a)

Grain of Interest



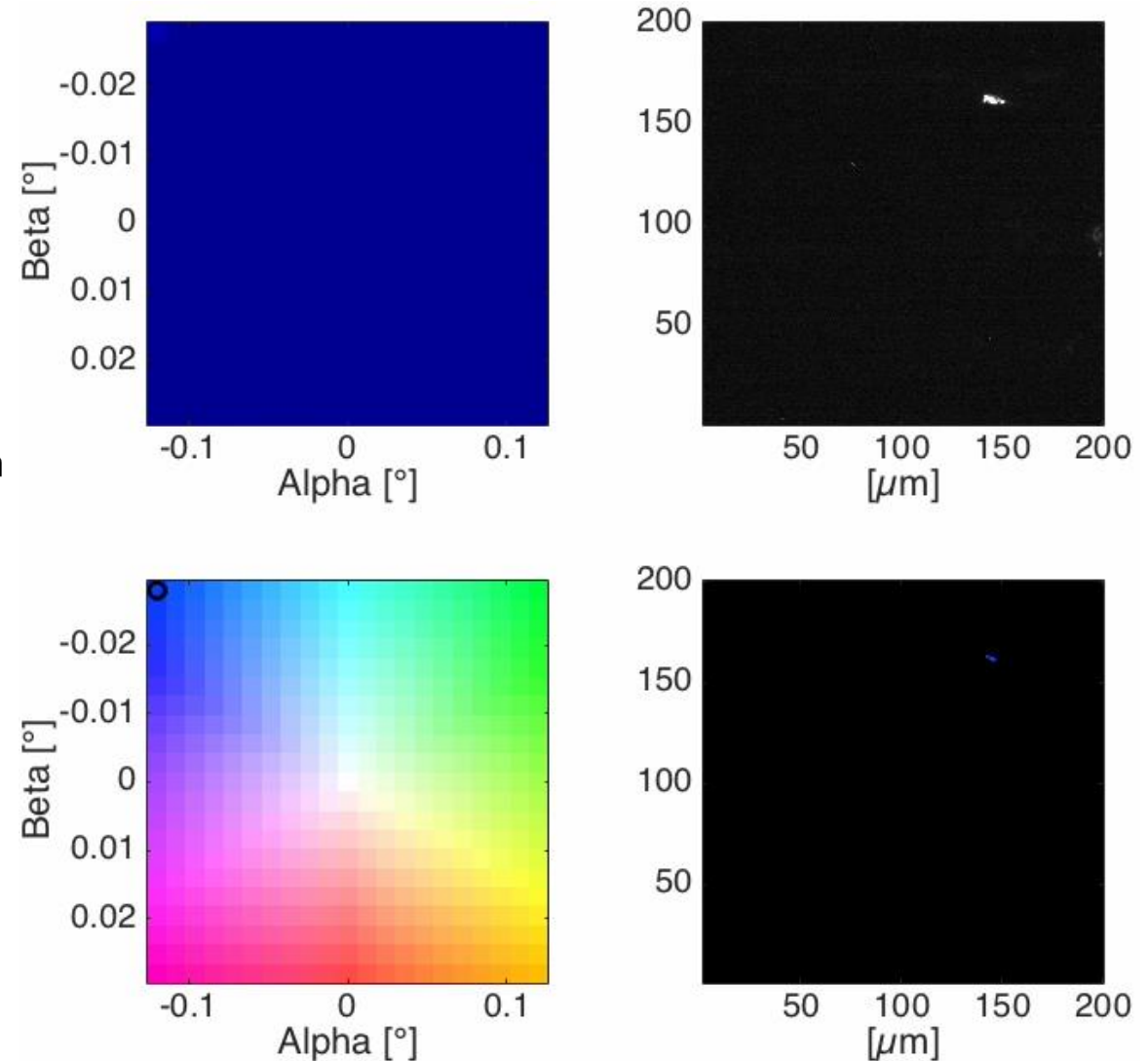
(b)

Dislocation Structures



(c)

Multi-scale Orientation/Strain Mapping from mm to nm range



Key parameters:

Working energy:

- typically 15-20 keV
- option for 33 keV

Focal length of the objective ~280 mm

Resolution:

- In practice ~150 nm
- Diffraction limit ~60 nm

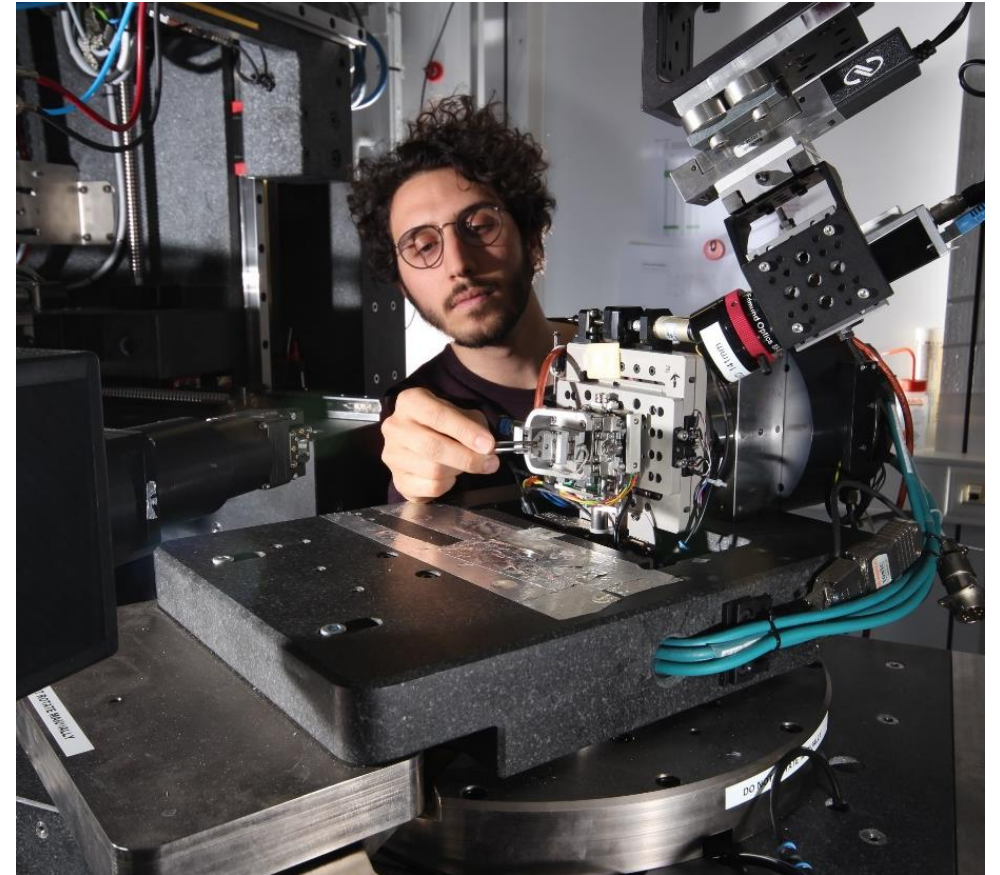
Multi-scale operation:

(we are still working on integrating this)

- 3DXRD for identification of grains of interest
- DCT for identification of grains of interest
- Near field camera (~1-2 μm pixel size)

2D \rightarrow 3D:

- Layer by layer with line beam
- Topo-tomo (in progress)



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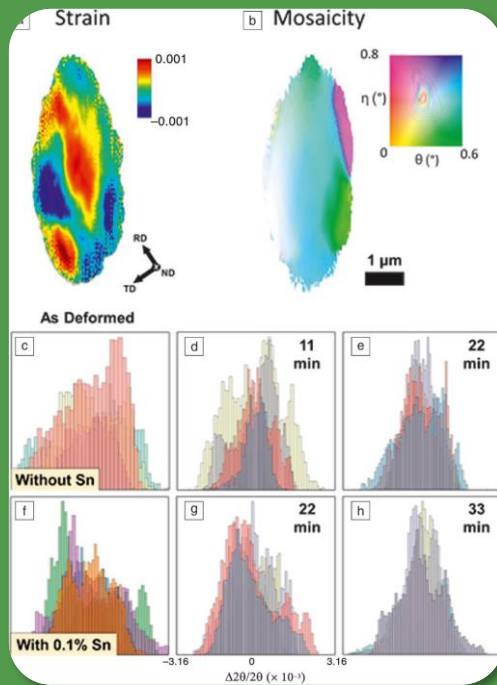
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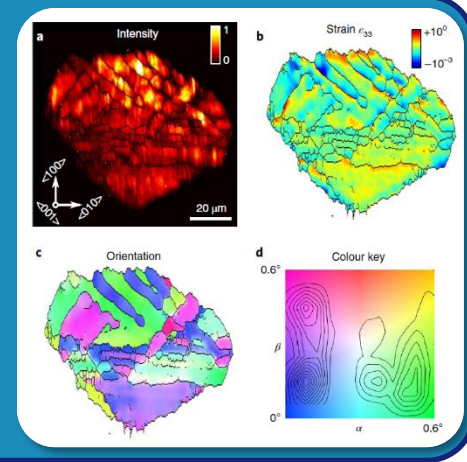
Metallurgy

- Pattern formation
- Materials fatigue
- Recovery and recrystallization



Functional materials

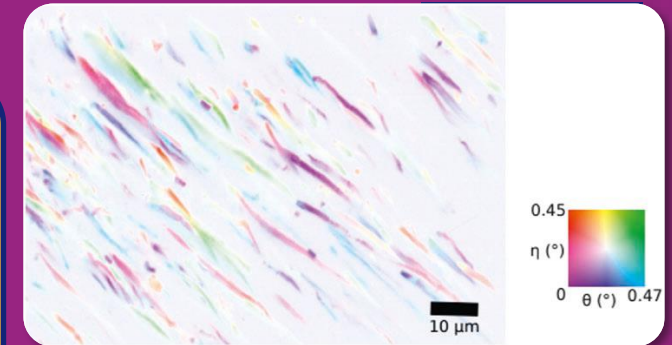
- Strain at grain boundaries
- Formation of domain patterns
- Dynamics of domain switching



3D strain maps with 150 nm spatial resolution

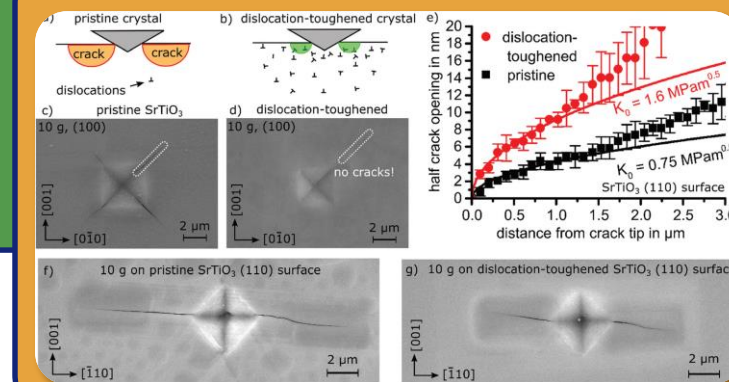
Biomaterials

- Microstructure



Ceramics

- Dislocation toughening
- Nano-twinning



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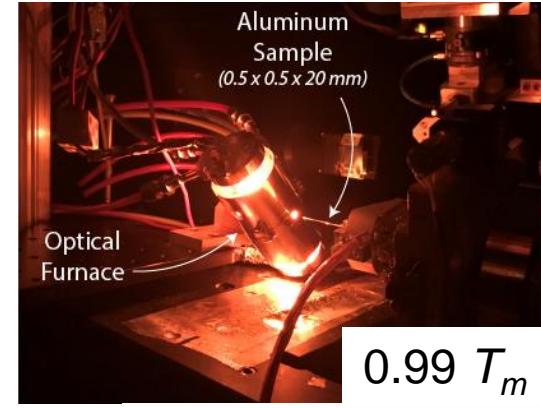
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TIME RESOLVED THERMALLY DRIVEN DISLOCATION MECHANICS

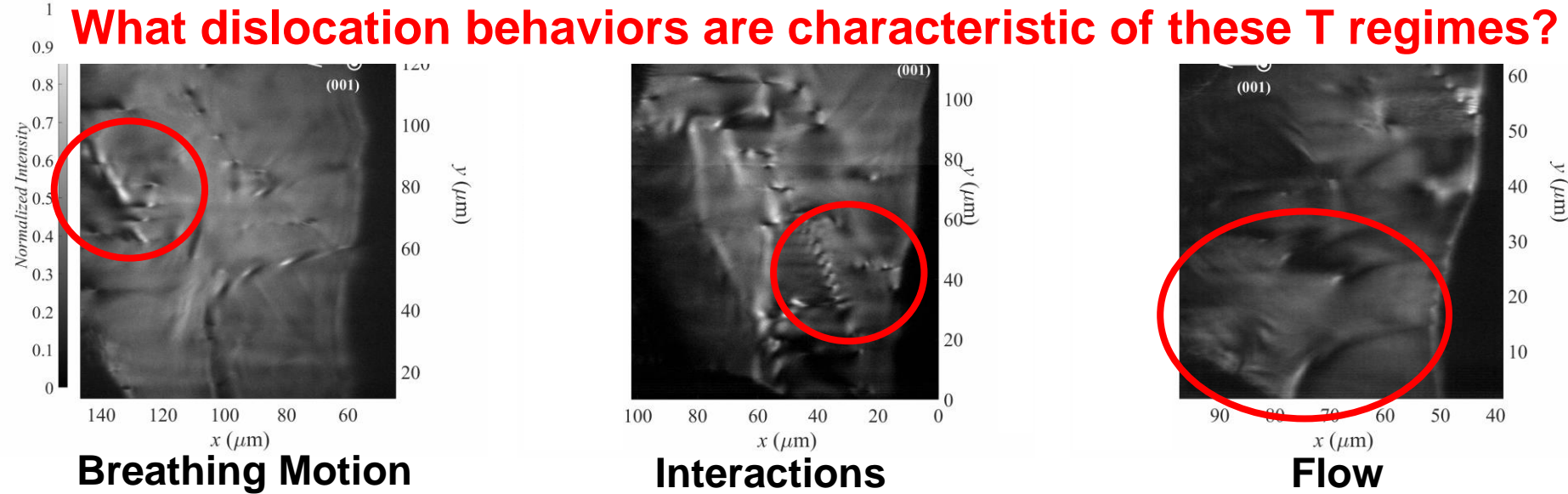
Thresholded Dynamics of Dislocations as Mobility Increases, $\sim 200\text{-}\mu\text{m}$ Below the Surface in Al single crystal



$0.93 T_m$
 $T = 606\text{ }^\circ\text{C}$
 $t = 0\text{ s}$

$0.97 T_m$
 $T = 627\text{ }^\circ\text{C}$
 $t = 0\text{ s}$

$0.99 T_m$
 $T = 646\text{ }^\circ\text{C}$
 $t = 0\text{ s}$



13

Dresselhaus Marais et al. Science Advances 2021 – Yildirim et al. Rev. Sci Inst. 2020

This work was performed under the auspices of the U.S. Department of Energy by Lawrence Livermore National Laboratory under Contract DE-AC52-07NA27344.

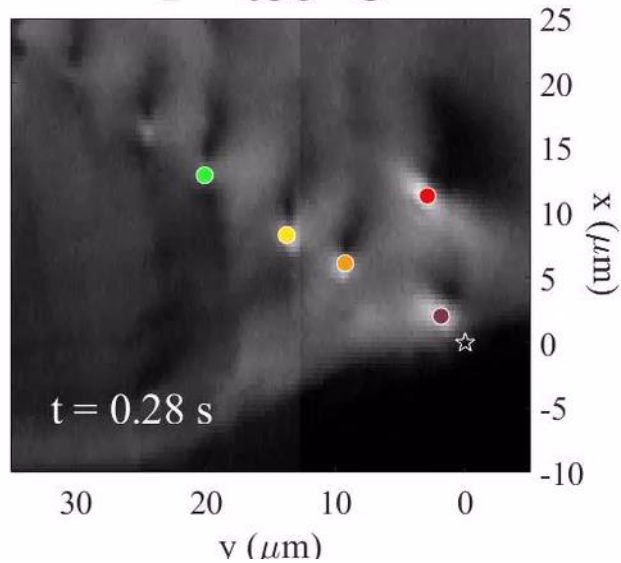
A NEW REGIME OF DISLOCATION BEHAVIOR

LLNL-PRES-818088



New View of Long-Range Interactions in Dislocation Structures

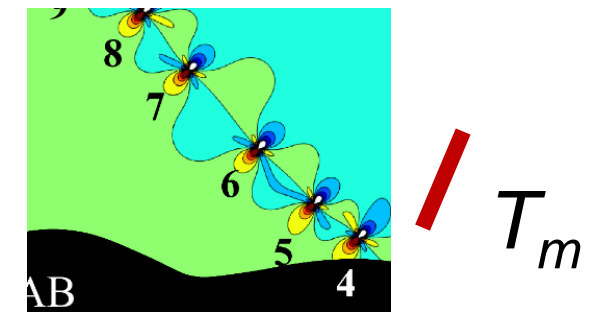
Classical Dislocation Dynamics
 $T = 638 \text{ }^\circ\text{C}$



Thermal Forces Amplify

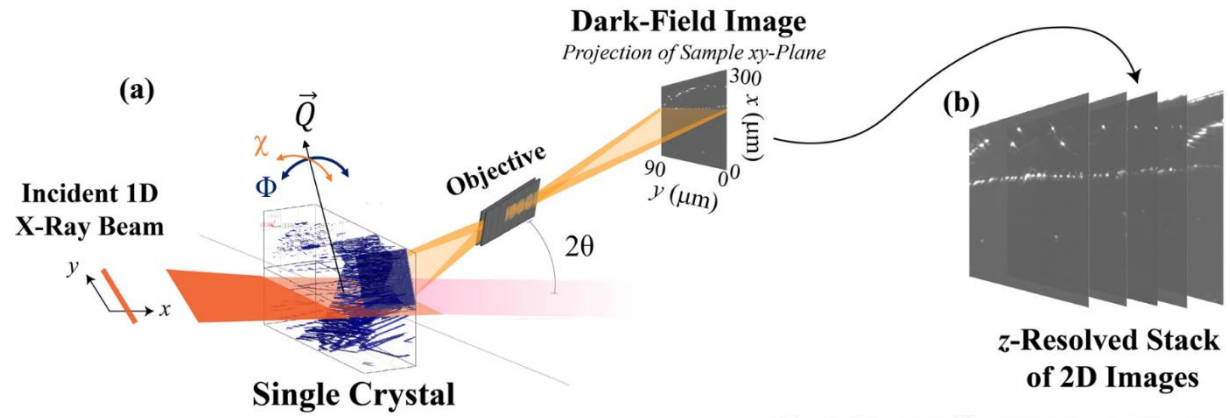
Thermal Effects Dominate Interactions

Insight into Regimes that Models & Experiments Struggle to Access



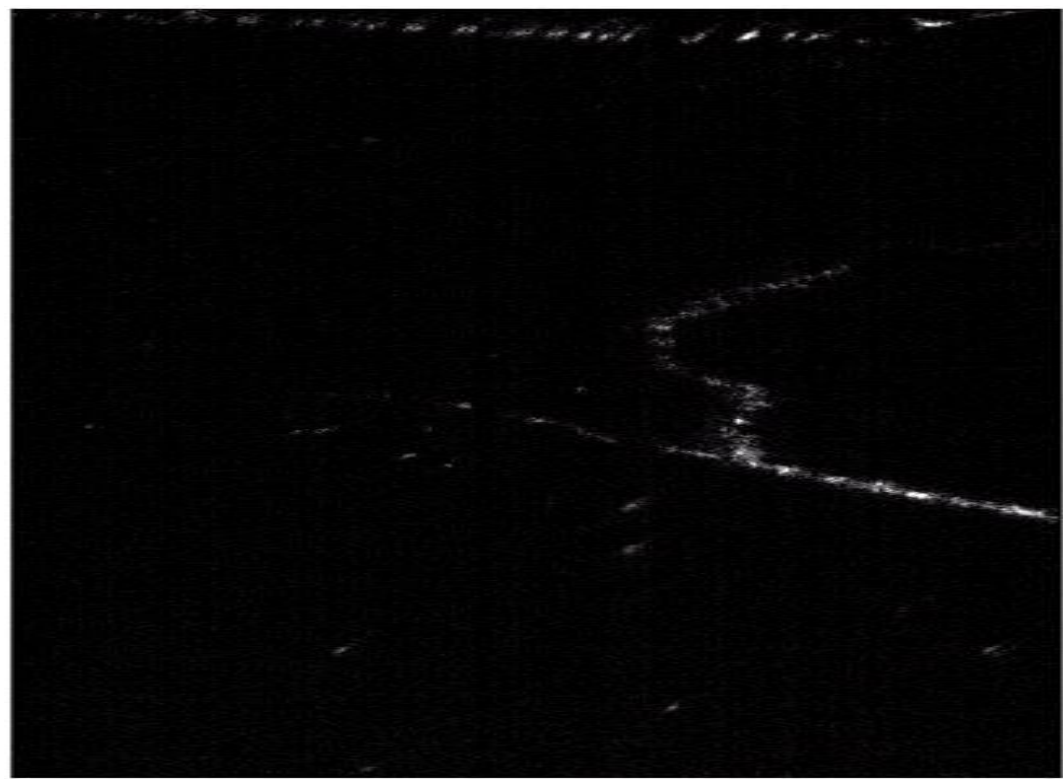
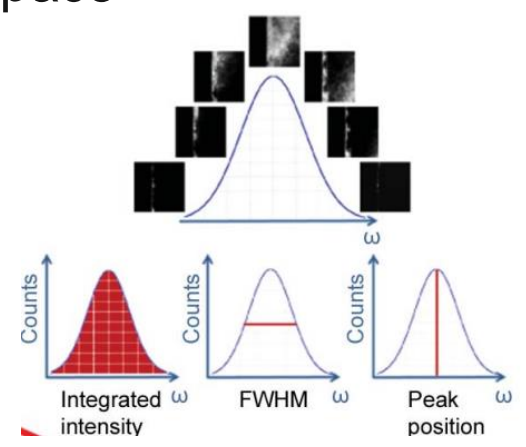
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DISLOCATIONS IN ANNEALED AL SINGLE CRYSTAL



Al Crystal annealed
10h at 590C

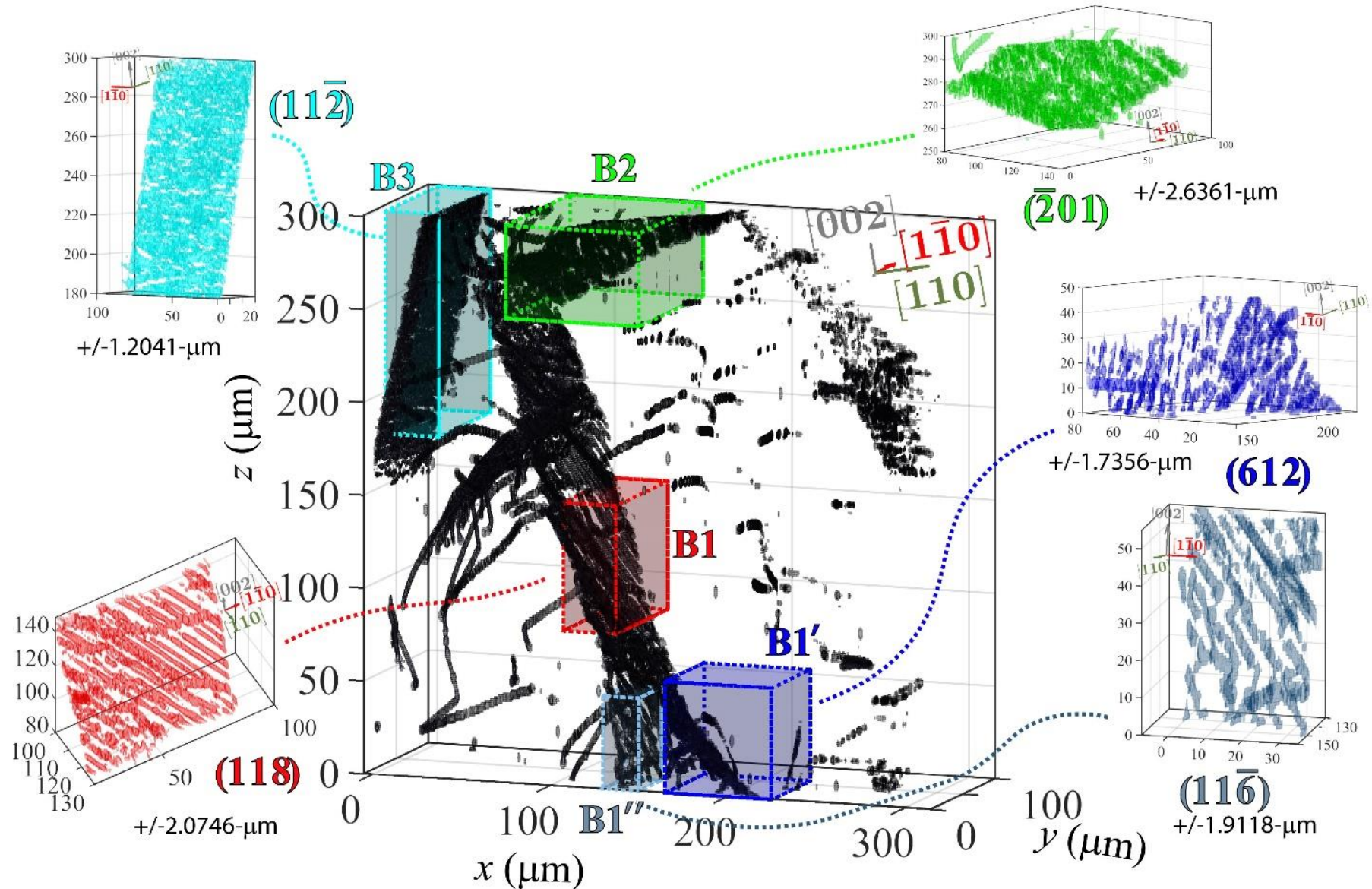
Weak-beam Z scans
shows dislocations in 3D
space



J. Appl. Cryst. (2017). 50, 561–569c

C. Yildirim, H. Poulsen, G. Winther, C. Detlefs,
P. Huang and L. Dresselhaus-Marais, *Sci Reports* 13, 3834 (2023).

SELF ORGANIZATION OF DISLOCATIONS INTO BOUNDARIES



C. Yildirim, H. Poulsen, G. Winther, C. Detlefs,
P. Huang and L. Dresselhaus-Marais, *Sci. Reports* 13, 3834 (2023)

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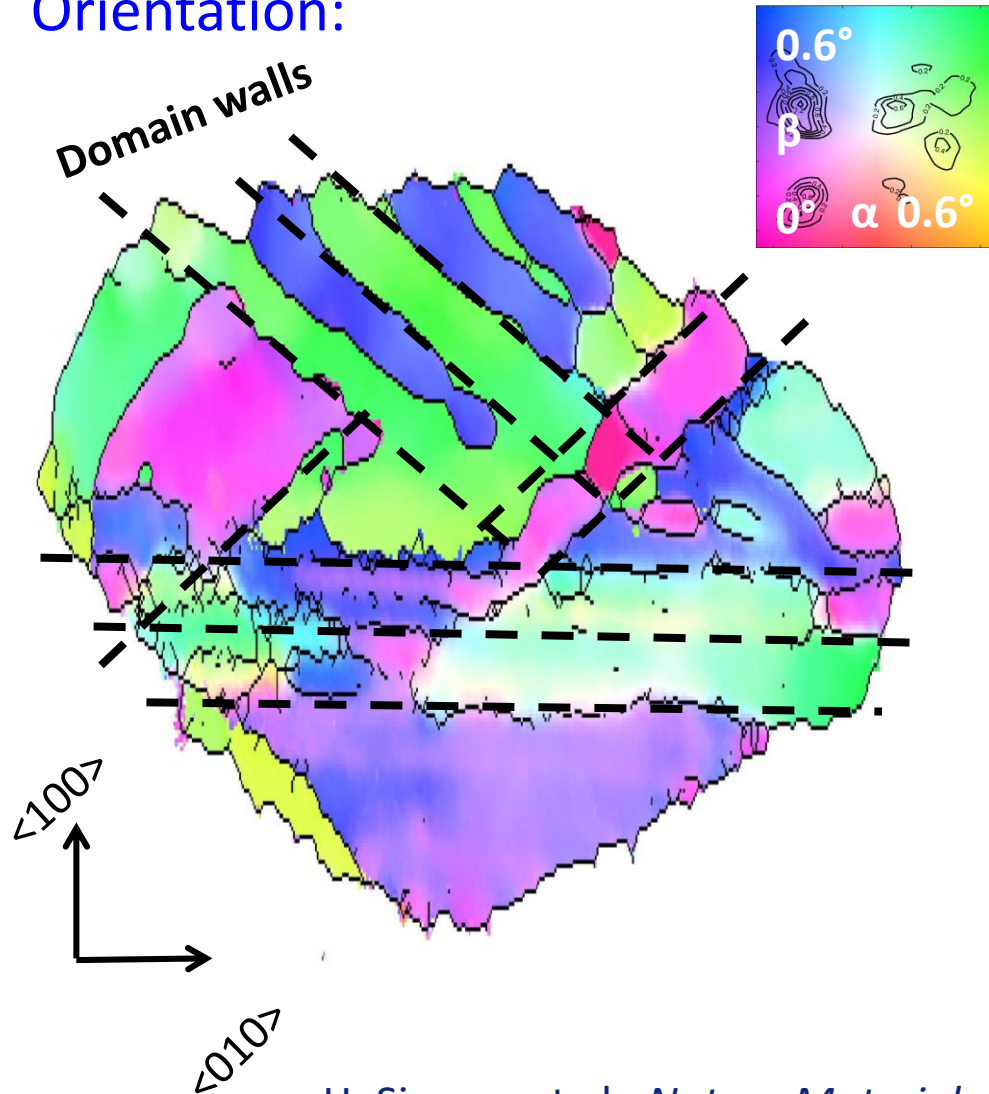
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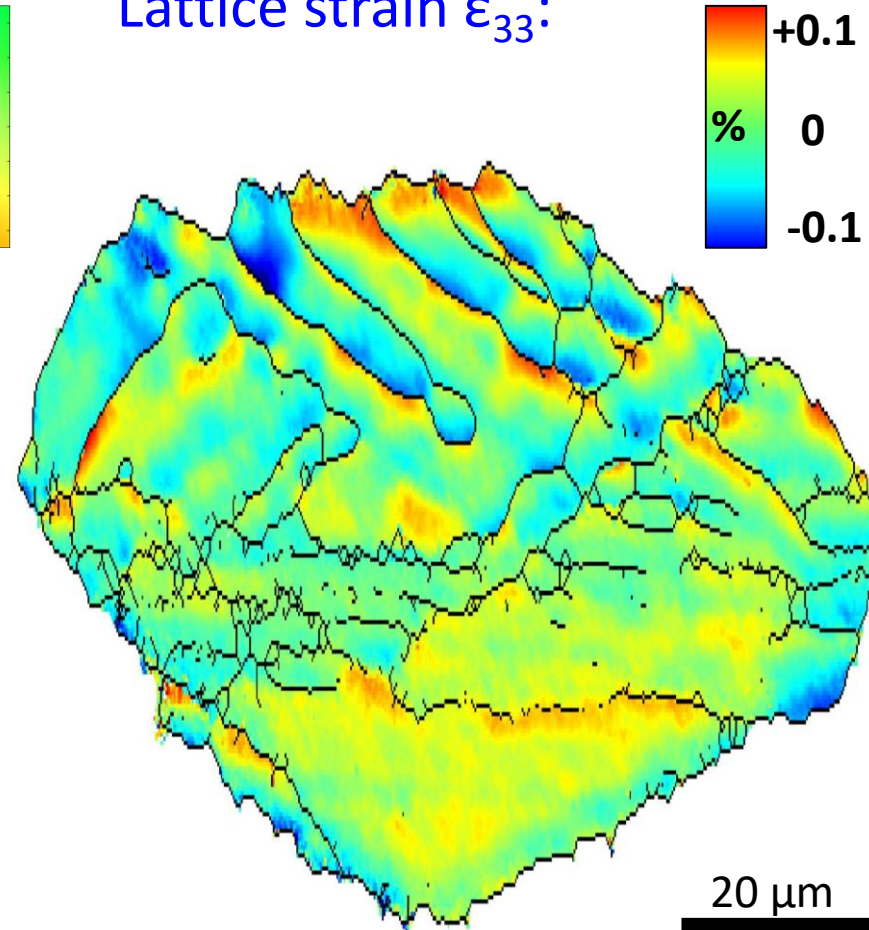
DOI:10.1557/mrs.2020.89

DOMAINS IN BaTiO₃

Orientation:



Lattice strain ϵ_{33} :



H. Simons et al., *Nature Materials* **17**, 815 (2018).

DOMAINS IN BaTiO₃

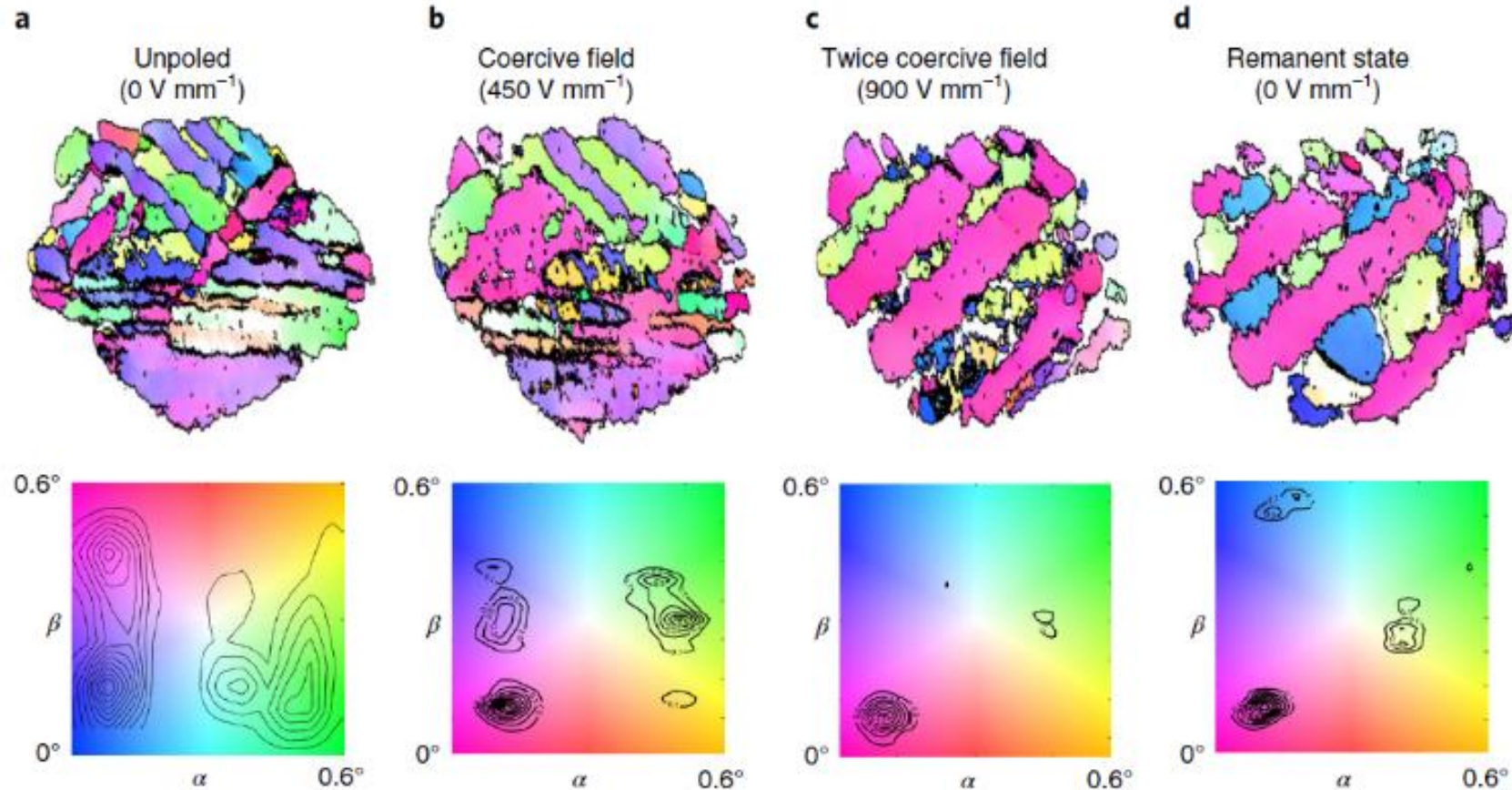


Fig. 4 | Changes to the domain topology and orientation distribution in the embedded BaTiO₃ grain during the in situ application of a unipolar electric field cycle along the $\langle 100 \rangle$ direction. **a-d**, Orientation maps (top) and diffraction intensity distributions (bottom) are shown for four points on a characteristic polarization (P) versus applied electric field (E) hysteresis curve: the initial zero-field state (**a**); at the coercive field, where most domain switching occurs (**b**); at twice the coercive field, where the induced polarization is saturated (**c**); and in the remanent state after the removal of the electric field altogether (**d**). Movies of the domain evolution as a function of field are available in the Supplementary Information.

H. Simons et al., *Nature Materials* **17**, 815 (2018).

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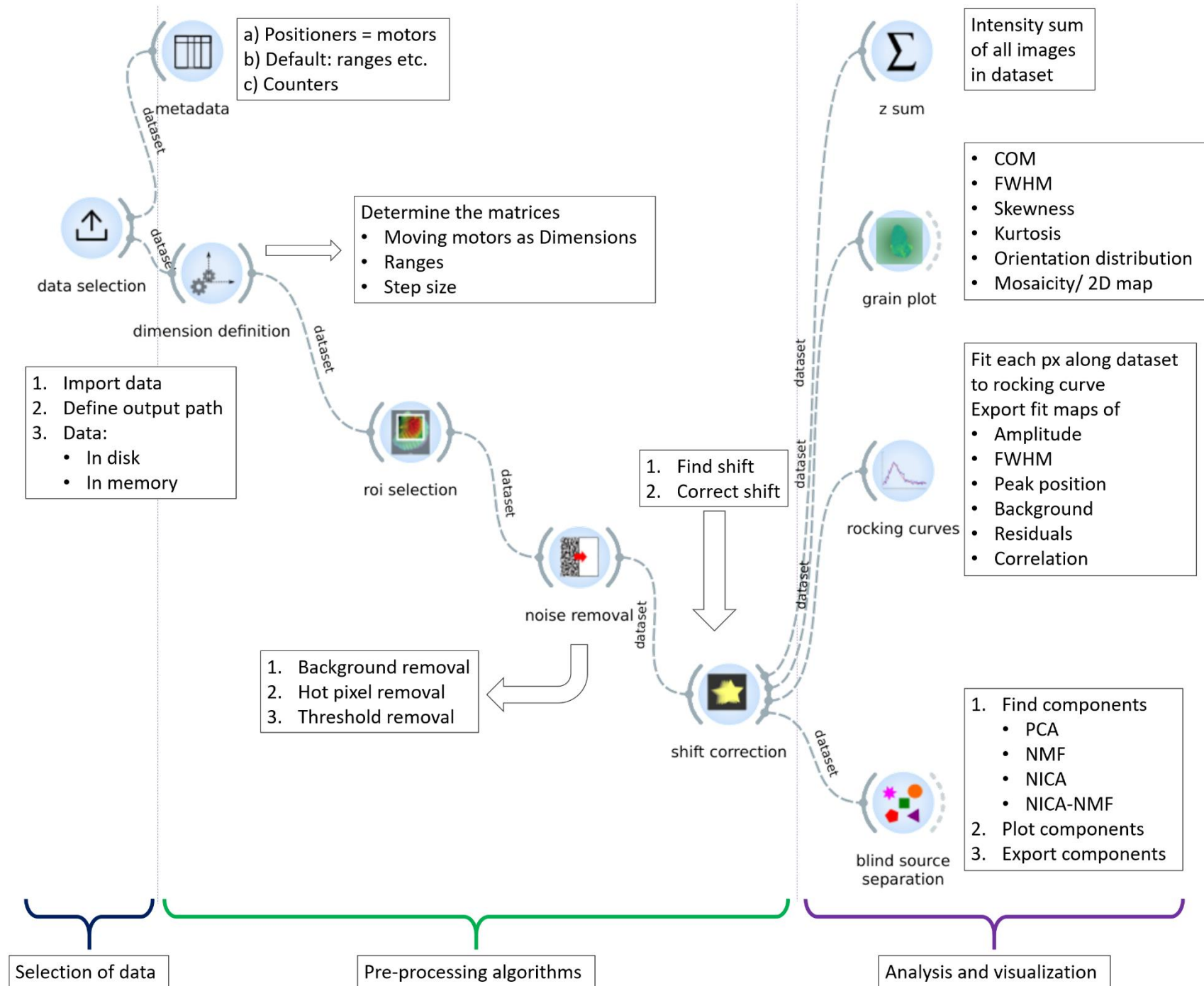
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Data analysis software

- add-on to Orange 3
- Widgets for all essential steps
 - Data selection,
 - Pre-processing
 - Processing (CCP4)
 - Blind source separation



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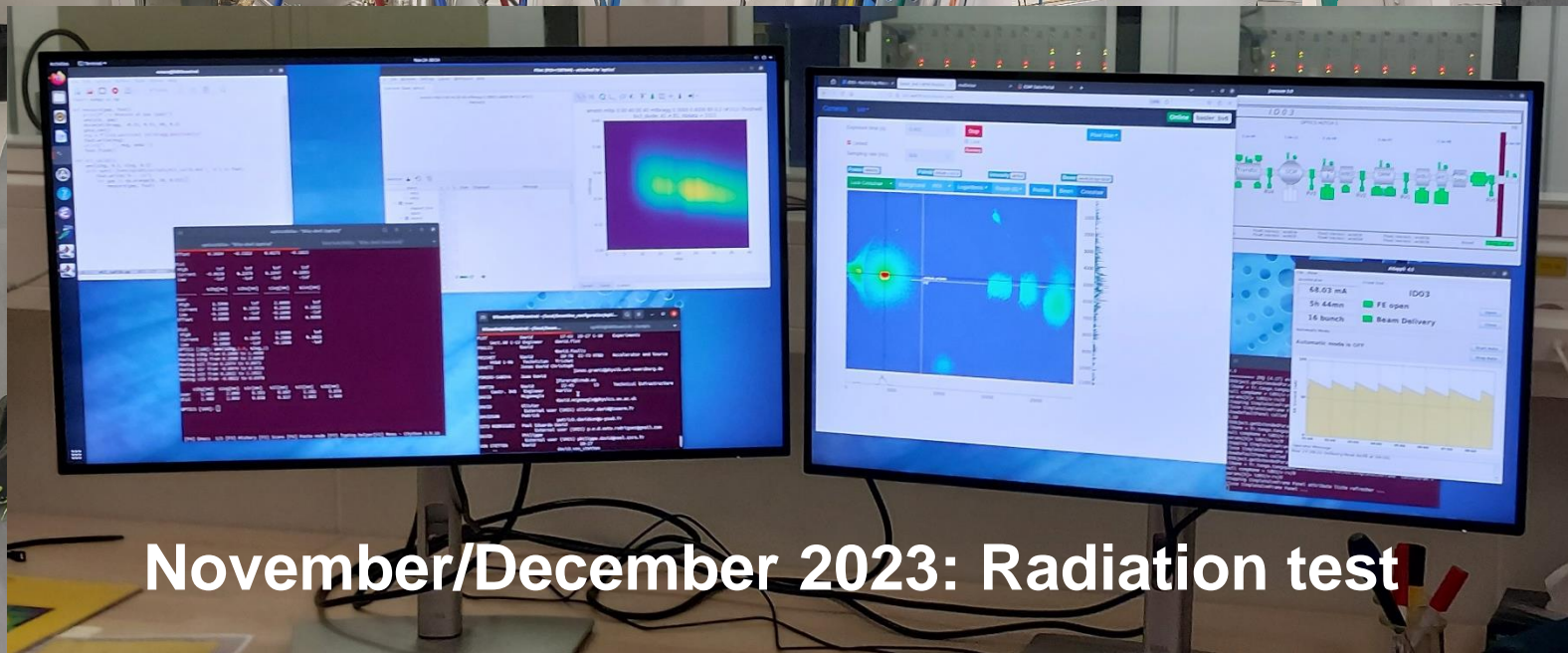
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ID03 replaces ID06-HXM

- Re-use the DFXM end-station from ID06-HXM
 - Rebuild the beamline from the ground up
 - Build new, fully optimized optics
-
- New, latest generation cryogenic undulator (CPMU)
 - Pink beam option, x100 flux
 - New sample goniometer optimized for topo-tomo
 - The beamline is fully dedicated to DFXM, therefore x2 more beamtime available



22



November/December 2023: Radiation test



November 2023



November 2023

Timeline

