

# Peculiarities of Development of High-Coercivity State of Sm-Co-Fe-Cu-Zr Magnets upon Step Cooling

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The Sm-Co-Fe-Cu-Zr magnets have the highest known maximum energy product  $(BH)_m$  together with high thermal stability of magnetic hysteresis properties. The coercivity  $H_c$  originates from the pinning of domain walls at interfaces of the  $\text{Sm}_2(\text{Co,Fe})_{17}$  ( $R3m - 2:17$ ) and  $\text{Sm}(\text{Co,Cu})_5$  ( $P6/mmm - 1:5$ ) phases of the nanocrystalline cellular structure, which is formed via the precipitation of the high-temperature-solid solution at 800-850°C [1]. The subsequent re-distribution of the alloy components between the 2:17 and 1:5 phases in the course of step cooling from 800 to 400°C increases coercivity  $H_c$  of the magnets. This work studies with XRD analysis of peculiarities of high-coercivity state development in the high temperature (HTPM -  $\text{Sm}(\text{Co}_{0.76}\text{Fe}_{0.12}\text{Cu}_{0.09}\text{Zr}_{0.03})_{7.0}$ ) and high energy (HEPM -  $\text{Sm}(\text{Co}_{0.64}\text{Fe}_{0.28}\text{Cu}_{0.06}\text{Zr}_{0.02})_{7.57}$ ) permanent magnets in the course of the step annealing. The main increase of  $H_c$  in the HEPM (which has about 15 vol. % of the 1:5 phase) occurs after annealing at  $T = 700^\circ\text{C}$  (Fig. 1a). The quenching of the magnet from 830°C results in a large misfit of the  $c$  lattice parameters of the coherent 2:17 and 1:5 phases, as well as increased stresses at the interface. The Cu diffusion from the 2:17 into the 1:5 phase upon annealing at  $T = 700^\circ\text{C}$  leads to the relaxation of the stresses (Fig.1 d). At  $T < 700^\circ\text{C}$ ,  $H_c$  grows negligibly (Fig. 1a). The volume fraction of the 1:5 phase in the HTPM quenched from 830°C reaches 50%; and the difference in the  $c$  lattice parameters of the 2:17 and 1:5 phases is negligible (Fig.1 e). Therefore, the Cu redistribution at  $T = 700$  and 600°C is slow and insignificant. The annealing at 500°C considerably accelerates the Cu diffusion and increase of  $H_c$ , especially, this is true for the initial step of annealing ( $t < 0.2$  h). As is shown in [1], interface stresses grow upon annealing at the temperature lower than the Curie temperature of the 1:5 phase ( $\sim 550^\circ\text{C}$ ). The lattice extension of the 1:5 along the  $c$  axis originating from the magnetic elasticity increases the misfits (Fig.1 f) and stresses. The Cu-enrichment of the interface of the 2:17 and 1:5 phases relaxes the stresses and increases the boundary-energy gradient, as well as  $H_c$ .

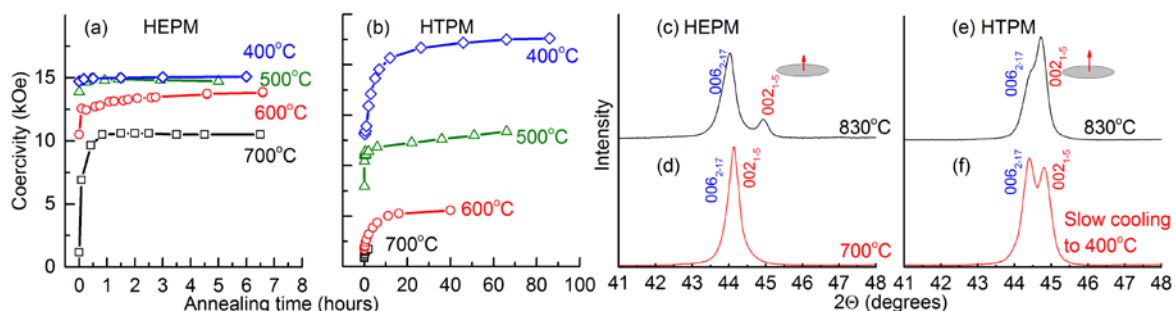


Figure 1: Coercivity kinetics (a,b) and fragments of x-ray diffraction patterns taken from the planes perpendicular to the texture direction of the HEPM (c, d) and the HTPM (e, f).

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## References

[1] - A. G. Popov, O. A. Golovnia, A. V. Protasov, et al. *IEEE Trans Magn* **54**, 6 (2018).